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(54) **LASER RE-MELT REPAIR OF
SUPERALLOYS USING FLUX**

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F01D 5/005; C23C 24/106; C23G 1/32
See application file for complete search history.

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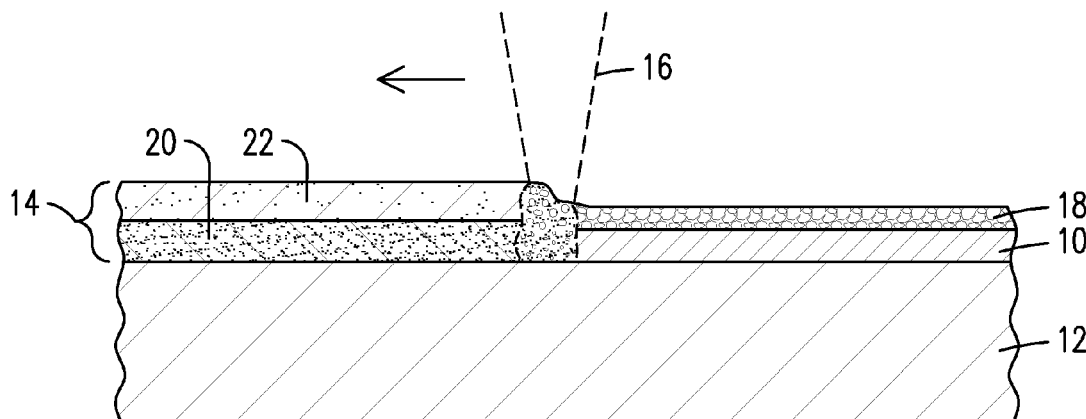
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Primary Examiner — Geoffrey S Evans

(57) **ABSTRACT**

A method of repairing service-induced surface cracks (92) in
a superalloy component (90). A layer of powdered flux mate-
rial (100) is applied over the cracks and is melted with a laser
beam (98) to form a re-melted zone (104) of the superalloy
material under a layer of slag (106). The slag cleanses the melt
pool of contaminants that may have been trapped in the
cracks, thereby eliminating the need for pre-melting fluoride
ion cleaning. Optionally, alloy feed material may be applied
with the powdered flux material to augment the volume of the
melt or to modify the composition of the re-melted zone.

18 Claims, 3 Drawing Sheets



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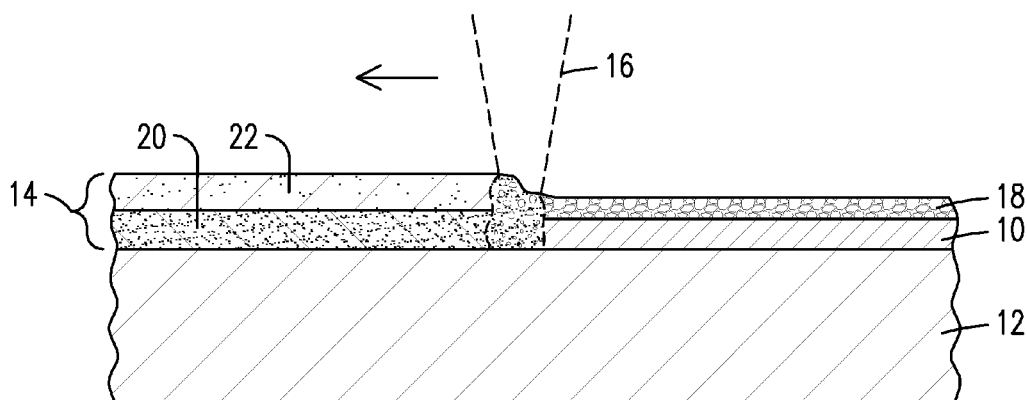


FIG. 1

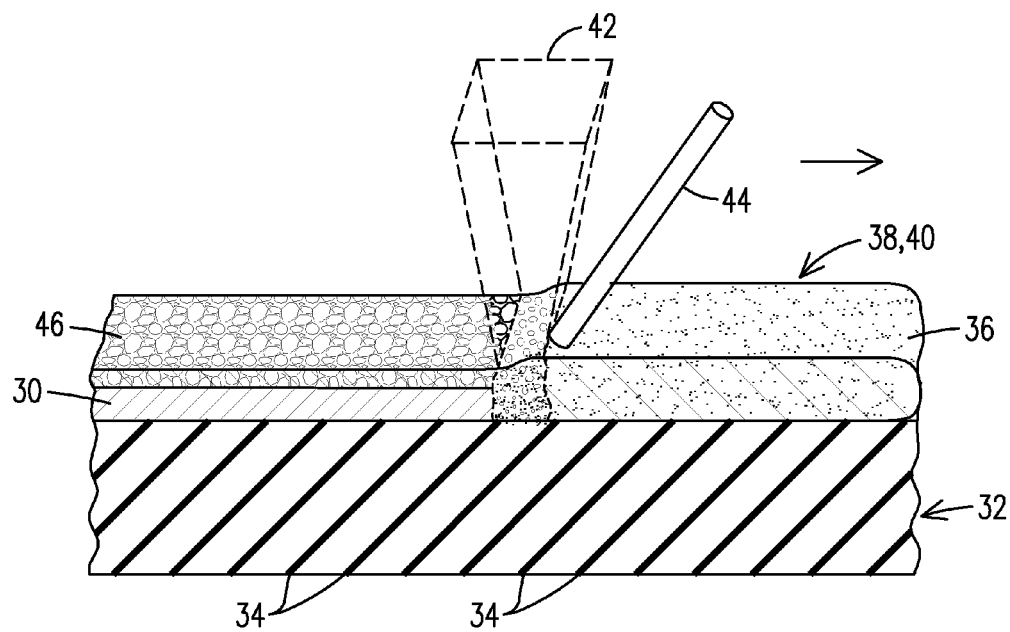


FIG. 2

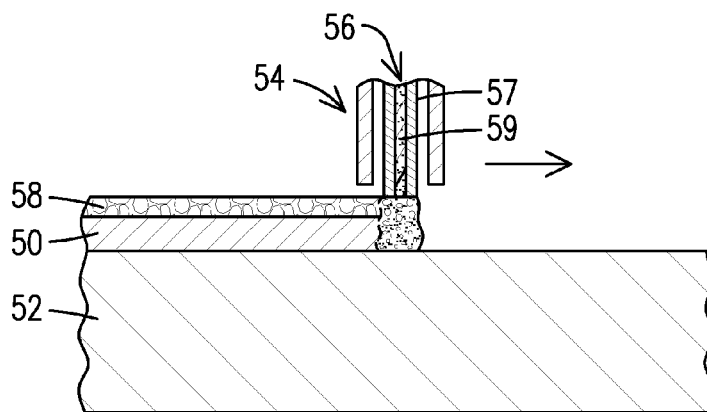


FIG. 3

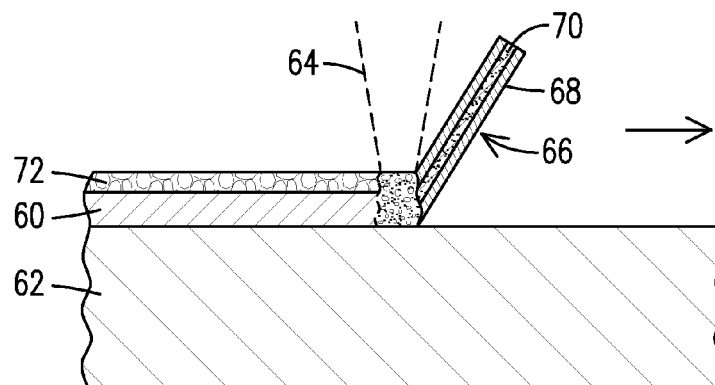


FIG. 4

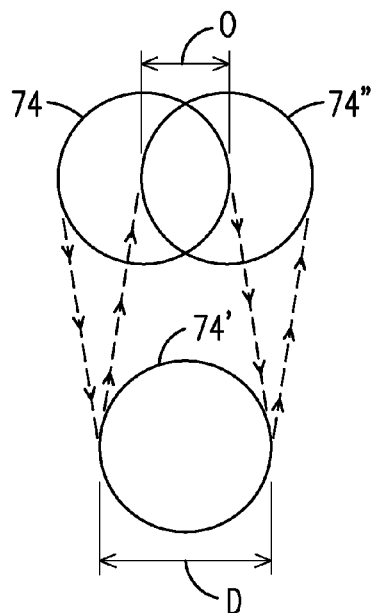


FIG. 5

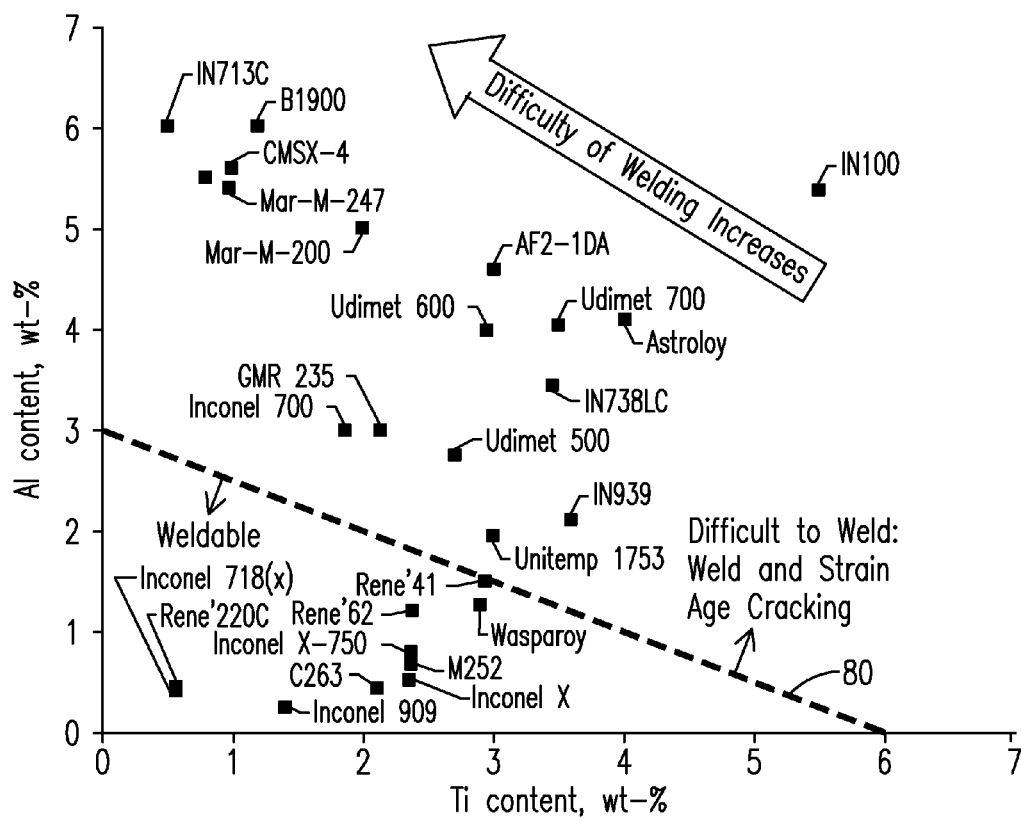


FIG. 6
PRIOR ART

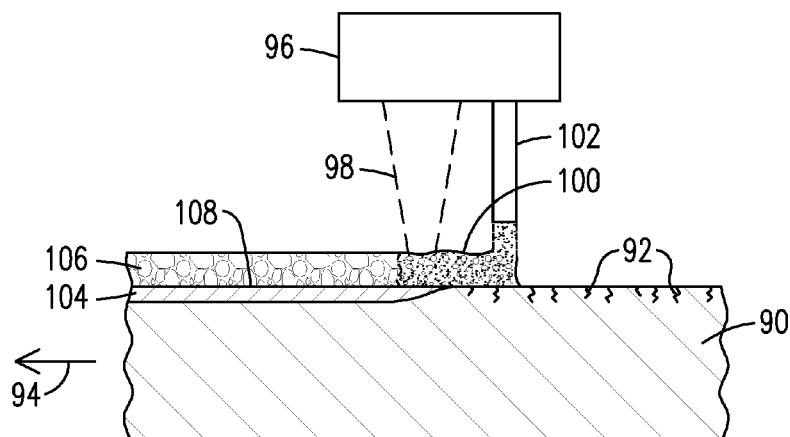


FIG. 7

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LASER RE-MELT REPAIR OF SUPERALLOYS USING FLUX

This application is a continuation-in-part of pending U.S. patent application Ser. No. 13/005,656 filed on 13 Jan. 2011 (publication number US 2012/0181255 A1), incorporated by reference herein.

FIELD OF THE INVENTION

This invention relates generally to the field of metals joining, and more particularly to the repair of superalloy materials.

BACKGROUND OF THE INVENTION

After service in a gas turbine engine, hot gas path components such as blades and vanes often develop shallow spider-like cracks known as craze cracking. Repair of such cracks is difficult because of the limited weldability of the superalloy materials of construction of such components. Superalloy materials are prone to weld and weld heat affected zone cracking during weld solidification and to reheat cracking upon subsequent post weld heat treatment that is required to re-establish material properties.

U.S. Pat. No. 7,169,242 describes a method for repairing small surface casting defects in superalloy materials by re-melting a 500 μ m deep surface layer of the material with a laser beam or other energy source under an inert atmosphere or vacuum. Preheating the article is described as allowing for the repair of deeper defects and for reducing the risk of hot tearing defects.

After-service cracks are typically contaminated with various combustion process products which make weld repair more difficult. Fluoride ion cleaning (FIC) can be used to remove contamination prior to welding, but it is of limited effectiveness for very narrow cracks, and it can be detrimental if applied too aggressively because of the potential for leaching out of important elements from the surrounding substrate material.

Welding processes vary considerably depending upon the type of material being welded. Some materials are more easily welded under a variety of conditions, while other materials require special processes in order to achieve a structurally sound joint without degrading the surrounding substrate material.

Common arc welding generally utilizes a consumable electrode as the feed material. In order to provide protection from the atmosphere for the molten material in the weld pool, an inert cover gas or a flux material may be used when welding many alloys including, e.g. steels, stainless steels, and nickel based alloys. Inert and combined inert and active gas processes include gas tungsten arc welding (GTAW) (also known as tungsten inert gas (TIG)) and gas metal arc welding (GMAW) (also known as metal inert gas (MIG) and metal active gas (MAG)). Flux protected processes include submerged arc welding (SAW) where flux is commonly fed, flux cored arc welding (FCAW) where the flux is included in the core of the electrode and shielded metal arc welding (SMAW) where the flux is coated on the outside of the filler electrode.

The use of energy beams as a heat source for welding is also known. For example, laser energy has been used to melt pre-placed stainless steel powder onto a carbon steel substrate with powdered flux material providing shielding of the melt pool. The flux powder may be mixed with the stainless steel

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powder or applied as a separate covering layer. To the knowledge of the inventors, flux materials have not been used when welding superalloy materials.

It is recognized that superalloy materials are among the most difficult materials to weld due to their susceptibility to weld solidification cracking and strain age cracking. The term "superalloy" is used herein as it is commonly used in the art; i.e., a highly corrosion and oxidation resistant alloy that exhibits excellent mechanical strength and resistance to creep at high temperatures. Superalloys typically include a high nickel or cobalt content. Examples of superalloys include alloys sold under the trademarks and brand names Hastelloy, Inconel alloys (e.g. IN 738, IN 792, IN 939), Rene alloys (e.g. Rene N5, Rene 80, Rene 142), Haynes alloys, Mar M, CM 247, CM 247 LC, C263, 718, X-750, ECY 768, 282, X45, PWA 1483 and CMSX (e.g. CMSX-4) single crystal alloys.

Weld repair of some superalloy materials has been accomplished successfully by preheating the material to a very high temperature (for example to above 1600° F. or 870° C.) in order to significantly increase the ductility of the material during the repair. This technique is referred to as hot box welding or superalloy welding at elevated temperature (SWET) weld repair, and it is commonly accomplished using a manual GTAW process. However, hot box welding is limited by the difficulty of maintaining a uniform component process surface temperature and the difficulty of maintaining complete inert gas shielding, as well as by physical difficulties imposed on the operator working in the proximity of a component at such extreme temperatures.

Some superalloy material welding applications can be performed using a chill plate to limit the heating of the substrate material; thereby limiting the occurrence of substrate heat affects and stresses causing cracking problems. However, this technique is not practical for many repair applications where the geometry of the parts does not facilitate the use of a chill plate.

FIG. 6 is a conventional chart illustrating the relative weldability of various alloys as a function of their aluminum and titanium content. Alloys such as Inconel® 718 which have relatively lower concentrations of these elements, and consequently relatively lower gamma prime content, are considered relatively weldable, although such welding is generally limited to low stress regions of a component. Alloys such as Inconel® 939 which have relatively higher concentrations of these elements are generally not considered to be weldable, or can be welded only with the special procedures discussed above which increase the temperature/ductility of the material and which minimize the heat input of the process. For purposes of discussion herein, a dashed line **80** indicates a border between a zone of weldability below the line **80** and a zone of non-weldability above the line **80**. The line **80** intersects 3 wt. % aluminum on the vertical axis and 6 wt. % titanium on the horizontal axis. Within the zone of non-weldability, the alloys with the highest aluminum content are generally found to be the most difficult to weld.

It is also known to utilize selective laser melting (SLM) or selective laser sintering (SLS) to melt a thin layer of superalloy powder particles onto a superalloy substrate. The melt pool is shielded from the atmosphere by applying an inert gas, such as argon, during the laser heating. These processes tend to trap the oxides (e.g. aluminum and chromium oxides) that are adherent on the surface of the particles within the layer of deposited material, resulting in porosity, inclusions and other defects associated with the trapped oxides. Post process hot isostatic pressing (HIP) is often used to collapse these voids, inclusions and cracks in order to improve the properties of the deposited coating.

For some superalloy materials in the zone of non-weldability there is no known acceptable welding or repair process. Furthermore, as new and higher alloy content superalloys continue to be developed, the challenge to develop commercially feasible joining and repair processes for superalloy materials continues to grow.

BRIEF DESCRIPTION OF THE DRAWINGS

The invention is explained in the following description in view of the drawings that show:

FIG. 1 illustrates a cladding process using a multi-layer powder.

FIG. 2 illustrates a cladding process using a mixed layer powder.

FIG. 3 illustrates a cladding process using a cored filler wire and a cold metal arc welding torch.

FIG. 4 illustrates a cladding process using a cored filler wire and an energy beam.

FIG. 5 illustrates an energy beam overlap pattern.

FIG. 6 is a prior art chart illustrating the relative weldability of various superalloys.

FIG. 7 illustrates a laser re-melt repair process utilizing flux powder.

DETAILED DESCRIPTION OF THE INVENTION

It is noted for the convenience of the reader that FIGS. 1-5 herein illustrate various aspects and applications of the inventive technology described herein, and that the description of FIG. 7 below is particularly directed to the presently claimed use of the inventive technology for laser re-melt repair of superalloy materials.

The present inventors have developed a materials joining process that can be used successfully to join and/or repair the most difficult to weld superalloy materials. While flux materials have not previously been utilized when welding superalloy materials, embodiments of the inventive process advantageously apply a powdered flux material over a superalloy substrate during a melting and re-solidifying process. Some embodiments also utilize the precise energy input control capability of energy beam heating processes, such as laser beam heating. The powdered flux material is effective to provide beam energy trapping, impurity cleansing, atmospheric shielding, bead shaping, and cooling temperature control in order to accomplish crack-free joining of superalloy materials without the necessity for high temperature hot box welding or the use of a chill plate or the use of inert shielding gas. While various elements of the present invention have been known in the welding industry for decades, the present inventors have innovatively developed a combination of steps for a superalloy joining process that solves the long-standing problem of cracking of these materials.

FIG. 1 illustrates a process where a layer of cladding 10 of a superalloy material is being deposited onto a superalloy substrate material 12 at ambient room temperature without any preheating of the substrate material 12 or the use of a chill plate. The substrate material 12 may form part of a gas turbine engine blade, for example, and the cladding process may be part of a repair procedure in some embodiments. A layer of granulated powder 14 is preplaced on the substrate 12, and a laser beam 16 is traversed across the layer of powder 14 to melt the powder and to form the layer of cladding 10 covered by a layer of slag 18. The cladding 10 and slag 18 are formed from the layer of powder 14 which includes a layer of powdered superalloy material 20 covered by a layer of powdered flux material 22.

The flux material 22 and resultant layer of slag 18 provide a number of functions that are beneficial for preventing cracking of the cladding 10 and the underlying substrate material 12. First, they function to shield both the region of molten material and the solidified (but still hot) cladding material 10 from the atmosphere in the region downstream of the laser beam 16. The slag floats to the surface to separate the molten or hot metal from the atmosphere, and the flux may be formulated to produce a shielding gas in some embodiments, thereby avoiding or minimizing the use of expensive inert gas. Second, the slag 18 acts as a blanket that allows the solidified material to cool slowly and evenly, thereby reducing residual stresses that can contribute to post weld reheat or strain age cracking. Third, the slag 18 helps to shape the pool of molten metal to keep it close to a desired $\frac{1}{3}$ height/width ratio. Fourth, the flux material 22 provides a cleansing effect for removing trace impurities such as sulfur and phosphorus that contribute to weld solidification cracking. Such cleansing includes deoxidation of the metal powder. Because the flux powder is in intimate contact with the metal powder, it is especially effective in accomplishing this function. Finally, the flux material 22 may provide an energy absorption and trapping function to more effectively convert the laser beam 16 into heat energy, thus facilitating a precise control of heat input, such as within 1-2%, and a resultant tight control of material temperature during the process. Additionally, the flux may be formulated to compensate for loss of volatilized elements during processing or to actively contribute elements to the deposit that are not otherwise provided by the metal powder itself. Together, these process steps produce crack-free deposits of superalloy cladding on superalloy substrates at room temperature for materials that heretofore were believed only to be joinable with a hot box process or through the use of a chill plate.

FIG. 2 illustrates another embodiment where a layer of cladding 30 of a superalloy material is being deposited onto a superalloy substrate material 32, which in this embodiment is illustrated as a directionally solidified material having a plurality of columnar grains 34. In this embodiment, the layer of powder 36 is pre-placed or fed onto the surface of the substrate material 32 as a homogeneous layer including a mixture of both powdered alloy material 38 and powdered flux material 40. The layer of powder 36 may be one to several millimeters in thickness in some embodiments rather than the fraction of a millimeter typical with known selective laser melting and sintering processes. Typical powdered prior art flux materials have particle sizes ranging from 0.5-2 mm, for example. However, the powdered alloy material 38 may have a particle size range (mesh size range) of from 0.02-0.04 mm or 0.02-0.08 mm or other sub-range therein. This difference in mesh size range may work well in the embodiment of FIG. 1 where the materials constitute separate layers; however, in the embodiment of FIG. 2, it may be advantageous for the powdered alloy material 38 and the powdered flux material 40 to have overlapping mesh size ranges, or to have the same mesh size range in order to facilitate mixing and feeding of the powders and to provide improved flux coverage during the melting process.

The energy beam 42 in the embodiment of FIG. 2 is a diode laser beam having a generally rectangular cross-sectional shape, although other known types of energy beams may be used, such as electron beam, plasma beam, one or more circular laser beams, a scanned laser beam (scanned one, two or three dimensionally), an integrated laser beam, etc. The rectangular shape may be particularly advantageous for embodiments having a relatively large area to be clad, such as for repairing the tip of a gas turbine engine blade. The broad

area beam produced by a diode laser helps to reduce weld heat input, heat affected zone, dilution from the substrate and residual stresses, all of which reduce the tendency for the cracking effects normally associated with superalloy repair. Optical conditions and hardware optics used to generate a broad area laser exposure may include but are not limited to: defocusing of the laser beam; use of diode lasers that generate rectangular energy sources at focus; use of integrating optics such as segmented mirrors to generate rectangular energy sources at focus; scanning (rastering) of the laser beam in one or more dimensions; and the use of focusing optics of variable beam diameter (e.g. 0.5 mm at focus for fine detailed work varied to 2.0 mm at focus for less detailed work). The motion of the optics and/or substrate may be programmed as in a selective laser melting or sintering process to build a custom shape layer deposit. Advantages of this process over known laser melting or sintering processes include: high deposition rates and thick deposit in each processing layer; improved shielding that extends over the hot deposited metal without the need for inert gas; flux will enhance cleansing of the deposit of constituents that otherwise lead to solidification cracking; flux will enhance laser beam absorption and minimize reflection back to processing equipment; slag formation will shape and support the deposit, preserve heat and slow the cooling rate, thereby reducing residual stresses that otherwise contribute to strain age (reheat) cracking during post weld heat treatments; flux may compensate for elemental losses or add alloying elements; and powder and flux preplacement or feeding can efficiently be conducted selectively because the thickness of the deposit greatly reduces the time involved in total part building.

The embodiment of FIG. 2 also illustrates the use of a base alloy feed material **44** (alternatively referred to as a filler material). The feed material **44** may be in the form of a wire or strip that is fed or oscillated toward the substrate **32** and is melted by the energy beam **42** to contribute to the melt pool. If desired, the feed material may be preheated (e.g. electrically) to reduce overall energy required from the laser beam. While it is difficult or impossible to form some superalloy materials into wire or strip form, materials such as pure nickel or nickel-chromium or nickel-chromium-cobalt are readily available in those forms. In the embodiment of FIG. 2, the base alloy feed material **44**, powdered alloy material **38** and powdered flux material **40** are advantageously selected such that the layer of cladding material **30** has the composition of a desired superalloy material. The filler material may be only an extrudable subset of elements of a composition of elements defining a desired superalloy material, and the powdered metal material includes elements that complement the elements in the filler material to complete the composition of elements defining the desired superalloy material. The filler material and the powdered metal material are combined in the melt pool to form the repaired surface of desired superalloy material **30**. As in FIG. 1, the process produces a layer of slag **46** that protects, shapes and thermally insulates the layer of cladding material **30**.

FIG. 3 illustrates an embodiment where a layer of superalloy material **50** is deposited onto a superalloy substrate **52** using a cold metal arc welding torch **54**. The torch **54** is used to feed and to melt a filler material **56** having the form of a cored wire or strip material including a hollow metal sheath **57** filled with a powdered core material **59**. The powdered core material **59** may include powdered metal alloy and/or flux materials. Advantageously, the metal sheath **57** is formed of a material that can be conveniently formed into a hollow shape, such as nickel or nickel-chromium or nickel-chromium-cobalt, and the powdered material **59** is selected such

that a desired superalloy composition is formed when the filler material **56** is melted. The sheath contains sufficient nickel (or cobalt) to achieve the desired superalloy composition, thus the solid to solid ratio of sheath verses powdered core material may be maintained at a ratio of 3:2, for example. The heat of the arc melts the filler material **56** and forms a layer of the desired superalloy material **50** covered by a layer of slag **58**. Powdered flux material may be provided in the filler material **56** (for example 25% of the core volume) or it may be pre-placed or deposited onto the surface of the substrate **52** (not shown—see FIG. 2), or the electrode may be coated with flux material, or any combination of these alternatives. A supplemental powdered metal material may also be added to the melt pool (not shown—see FIGS. 1 and 2) by being pre-placed on the surface of the substrate **52** or by being directly fed into the melt pool during the step of melting. In various embodiments, the flux may be electrically conductive (electroslag) or not (submerged arc), and it may be chemically neutral or additive. As before, the filler material may be preheated to reduce process energy required—in this case from the cold metal arc torch. The use of flux would provide shielding thereby reducing or eliminating the need for inert or partially inert gas commonly required in the cold metal arc process.

FIG. 4 illustrates an embodiment where a layer of superalloy material **60** is deposited onto a superalloy substrate **62** using an energy beam such as laser beam **64** to melt a filler material **66**. As described above with respect to FIG. 3, the filler material **66** includes a metal sheath **68** that is constructed of a material that can be conveniently formed into a hollow shape, such as nickel or nickel-chromium or nickel-chromium-cobalt, and a powdered material **70** is selected such that a desired superalloy composition is formed when the filler material **66** is melted by the laser beam **64**. The powdered material **70** may include powdered flux as well as alloying elements. The heat of the laser beam **64** melts the filler material **66** and forms a layer of the desired superalloy material **60** covered by a layer of slag **72**. As before, the filler material may be preheated, such as with an electrical current, to reduce process energy required—in this case from the laser beam.

One embodiment of a filler material **56**, **66** is formulated to deposit alloy 247 material as follows:

sheath solid volume is about 60% of total metallic solid volume and is pure Ni;

core metal powder volume is about 40% of total metallic solid volume including sufficient Cr, Co, Mo, W, Al, Ti, Ta, C, B, Zr and Hf; that when melted together and mixed with the pure nickel from the sheath, produces alloy 247 composition of nominal weight percent 8.3 Cr, 10 Co, 0.7 Mo, 10 W, 5.5 Al, 1 Ti, 3 Ta, 0.14 C, 0.015 B, 0.05 Zr and 1.5 Hf; and

core flux powder volume represents additional, largely non-metallic, wire volume possibly about equal in size to the metal powder volume and includes alumina, fluorides and silicates in a 35/30/35 ratio. The mesh size range of the flux is such as to distribute uniformly within the core metal powder.

For embodiments where the heat of melting is provided by an arc, it is common to provide oxygen or carbon dioxide in the flux or shielding gas in order to maintain arc stability. However, the oxygen or carbon dioxide will react with titanium and some of the titanium will be lost as vapor or oxides during the melting process. The present invention allows the amount of titanium included in the filler material to be in excess of the amount of titanium desired in the deposited superalloy composition to compensate for this loss. For the

example of alloy 247 described above, the amount of titanium included in the core metal powder may be increased from 1% to 3%.

One will appreciate that other alloys, such as stainless steels for example, may be deposited with a similar process where a cored feed material is filled with a powdered core material including powdered flux and powdered metal. The powdered metal may be used to augment the composition of the sheath material to obtain a cladding material of a desired chemistry. For embodiments where there is a loss of material due to vaporization during the melting step, the powdered metal may include an excess of the lost material to compensate for the loss. For example, when alloy 321 stainless steel sheath material is deposited under a shielding gas containing oxygen or carbon dioxide, some of the titanium from the sheath material is lost due to reaction with the oxygen or carbon dioxide. The powdered core material in such an embodiment may include powdered flux and powdered titanium to compensate for the loss, thus providing a desired alloy 321 cladding composition.

Repair processes for superalloy materials may include preparing the superalloy material surface to be repaired by grinding as desired to remove defects, cleaning the surface, then pre-placing or feeding a layer of powdered material containing flux material onto the surface, then traversing an energy beam across the surface to melt the powder and an upper layer of the surface into a melt pool having a floating slag layer, then allowing the melt pool and slag to solidify. The melting functions to heal any surface defects at the surface of the substrate, leaving a renewed surface upon removal of the slag typically by known mechanical and/or chemical processes. The powdered material may be only flux material, or for embodiments where a layer of superalloy cladding material is desired, the powdered material may contain metal powder, either as a separate layer placed under a layer of powdered flux material, or mixed with the powdered flux material, or combined with the flux material into composite particles, such the melting forms the layer of cladding material on the surface. Optionally, a feed material may be introduced into the melt pool in the form of a strip or wire. The powdered metal and feed material (if any), as well as any metal contribution from the flux material which may be neutral or additive, are combined in the melt pool to produce a cladding layer having the composition of a desired superalloy material. In some embodiments, a feed material of nickel, nickel-chromium, nickel-chromium-cobalt or other conveniently extruded metal is combined with appropriate alloying metal powders to produce the desired superalloy composition in the cladding, thereby avoiding the problem of forming the desired superalloy material into a wire or strip form.

While pre-heating of the substrate is not necessarily required to obtain acceptable results, it may be desired to apply heat to the superalloy substrate and/or to the feed material and/or the powder prior to the melting step in some embodiments, such as to increase the ductility of the substrate material and/or to reduce beam energy otherwise required to melt the filler. Ductility improvement of some superalloy substrates is achieved at temperatures above about 80% of the alloy's melting point. Similarly, a chill fixture could optionally be used for particular applications, which in combination with the precision heat input of an energy beam can minimize stresses created in the material as a result of the melting process. Furthermore, the processes described herein may negate the need for an inert shielding gas, although supplemental shielding gas may be used in some applications if preferred. If a filler material 44 is used, it may be pre-heated in some embodiments.

Flux materials which could be used include commercially available fluxes such as those sold under the names Lincoln-weld P2007, Bohler Soudokay NiCrW-412, ESAB OK 10.16 or 10.90, Special Metals NT100, Oerlikon OP76, Sandvik 50SW or SAS1. The flux particles may be ground to a desired smaller mesh size range before use. Flux materials known in the art may typically include alumina, fluorides and silicates. Embodiments of the processes disclosed herein may advantageously include metallic constituents of the desired cladding material, for example chrome oxides, nickel oxides or titanium oxides. Any of the currently available iron, nickel or cobalt based superalloys that are routinely used for high temperature applications such as gas turbine engines may be joined, repaired or coated with the inventive process, including those alloys mentioned above.

Other variations may provide the heat for melting through the feed material rather than or in combination with an energy beam. For example, the wire or strip feed material 44 of FIG. 2 may be energized to create an arc under the layer of powder and flux, with the wire being a material that is readily available in extruded form (i.e. not a superalloy material) and the powder including the other alloying elements necessary to form a desired superalloy composition in the combined melt pool. Alternatively, the powder and flux may be selected to be conductive such as to facilitate an electro-slag welding process effective to form the layer of superalloy cladding material. In yet another embodiment, flux powder mixed with superalloy powder material may be fed to a superalloy substrate using conventional plasma arc cladding equipment, optionally with a chill fixture. The substrate, feed material and/or powder may be preheated in various embodiments. Because the degree of precision of the heat input is higher with the energy beam ($\pm 1-2\%$) than with an electrode ($\pm 10-15\%$), it may be desirable to utilize the energy beam for more than half of the total heat input. The beam energy may lead the submerged arc or electrosag process to initiate a preliminary melt pool with minimum dilution from the substrate, then the submerged arc or electrosag contribution can add to the volume of deposit without significant further substrate impact, thereby minimizing dilution effects.

In accordance with other embodiments, mixed submerged arc welding flux and alloy 247 powder was pre-placed from 2.5 to 5.5 mm depths and demonstrated to achieve crack free laser clad deposits after final post weld heat treatment. Ytterbium fiber laser power levels from 0.6 up to 2 kilowatts have been used with galvanometer scanning optics making deposits from 3 to 10 mm in width at travel speeds on the order of 125 mm/min. Absence of cracking has been confirmed by dye penetrant testing and metallographic examination of deposit cross sections. It will be appreciated that alloy 247 falls within the most difficult area of the zone of non-weldability as illustrated in FIG. 6, thereby demonstrating the operability of the invention for a full range of superalloy compositions, including those with aluminum content of greater than 3 wt. %.

It is appreciated that the advantages of utilizing powdered flux material when repairing a superalloy substrate are realized whether or not an additive cladding material is deposited. Surface cracks in a superalloy substrate may be repaired by covering the surface with powdered flux material, then heating the surface and the flux material to form a melt pool with a floating slag layer. Upon solidification of the melt pool under the protection of the slag layer, a clean surface with no cracks will be formed.

Laser energy may be applied across a surface area by using a diode laser having a generally rectangular energy density. Alternatively, it is possible to raster a circular laser beam back and forth as it is moved forward along a substrate to effect an

area energy distribution. FIG. 5 illustrates a rastering pattern for one embodiment where a generally circular beam having a diameter D is moved from a first position 74 to a second position 74' and then to a third position 74" and so on. An amount of overlap O of the beam diameter pattern at its locations of a change of direction is preferably between 25-90% of D in order to provide optimal heating and melting of the materials. Alternatively, two energy beams may be rastered concurrently to achieve a desired energy distribution across a surface area, with the overlap between the beam patterns being in the range of 25-90% of the diameters of the respective beams.

It will be appreciated that the use of powdered material facilitates the deposition of functionally graded materials, where the composition of the deposited material varies across time and space. For example, the alloy composition may vary from an interior wall to an exterior wall of a product, or from within a product to near its surfaces. The alloy composition may be varied in response to anticipated operating conditions requiring different mechanical or corrosion resistance properties, and with consideration of the cost of the materials.

FIG. 7 illustrates a process for the repair of near-surface defects, such as craze cracking of after service superalloy gas turbine hot gas path components. A substrate material 90 having one or more near-surface cracks 92 is moved in the direction of arrow 94 relative to a repair apparatus 96. An energy beam such as laser beam 98 is directed by the repair apparatus 96 toward a layer of powder 100 including powdered flux which is fed onto the surface through a nozzle 102 or which is optionally pre-placed on the substrate 90. The laser beam 98 melts the powder 100 and a thin surface region of the substrate 90 and forms a re-melted zone 104 in the substrate 90 covered by a layer of slag 106. As the re-melted zone 104 moves away from the laser beam 98 it re-solidifies under the slag 106 to form a repaired, crack-free surface 108. The flux of the powder 100 shields the melt and the hot trailing material from reaction with air without the necessity of using an inert cover gas. The flux is also effective to couple with and to trap the laser and heat energy in order to ensure melting sufficiently deep into the substrate 90 to consume the surface cracks 92.

The process of FIG. 7 advantageously provides a mechanism for cleansing and deoxidizing the melt pool, thereby removing contaminants that exist within the cracks 92 without the necessity for pre-weld fluoride ion cleaning by reacting with them and floating them to the surface to form part of the slag 106. The substrate 90 may be surface cleaned, such as with a wire brush or grit blasting process, in order to remove any overlying ceramic thermal barrier coating or surface adherent contamination prior to the re-melting process, but subsurface crack cleaning, such as fluoride ion cleaning, is optional and may be avoided in some embodiments. The process facilitates the repair of service-induced cracks in superalloy materials, including those materials having compositions in the zone of non-weldability above line 80 of FIG. 6 without the necessity for pre-heating and without post weld reheat cracking. While pre-heating of the substrate is not necessarily required to obtain acceptable results, it may be desired to apply heat to the superalloy substrate and/or the powder prior to the melting step in some embodiments, such as to increase the ductility of the substrate material and/or to reduce beam energy otherwise required.

The powder 100 of the process of FIG. 7 may optionally include some powdered alloy material as filler material, either pre-mixed and fed with the flux or fed or pre-placed separately to add to the re-melted zone. Alternatively, the filler material could be fed as wire or strip material as illustrated in

FIG. 2. Such filler may be of the same composition as the substrate 90 or it may be of a different composition, such as for providing somewhat different mechanical properties to the re-melted zone 104. As discussed above, the flux and filler powders may have overlapping or identical mesh size ranges. Various type lasers may be used (CO₂, NdYAG, fiber, diode, etc.) with various types of optics (focused, defocused, integrated beam, scanned beam, etc.).

While various embodiments of the present invention have been shown and described herein, it will be obvious that such embodiments are provided by way of example only. Numerous variations, changes and substitutions may be made without departing from the invention herein. Accordingly, it is intended that the invention be limited only by the spirit and scope of the appended claims.

The invention claimed is:

1. A method comprising:

applying powdered flux material to a surface of a superalloy substrate containing a surface opening defect;
traversing an energy beam across the surface to form a re-melted zone in the substrate covered by an overlying slag layer without previously performing any subsurface crack cleaning step; and
allowing the re-melted zone to solidify under the slag layer to form a repaired surface free of the defect.

2. The method of claim 1, wherein the energy beam is a laser beam.

3. The method of claim 1, further comprising applying a filler material to the surface during the step of traversing an energy beam such that melted filler material is additive to the re-melted zone.

4. The method of claim 3, further comprising applying the filler material to the surface as powdered alloy material.

5. The method of claim 4, wherein a mesh size range of the powdered alloy material overlaps with a mesh size range of the powdered flux material.

6. The method of claim 3, further comprising applying the filler material as wire or strip material.

7. The method of claim 1, wherein the superalloy substrate comprises a composition beyond a zone of weldability defined on a graph of superalloys plotting titanium content verses aluminum content, wherein the zone of weldability is upper bounded by a line intersecting the titanium content axis at 6 wt. % and intersecting the aluminum content axis at 3 wt. %.

8. The method of claim 1, further comprising applying heat to the substrate prior to the step of traversing the energy beam.

9. A method comprising:

covering a substrate with a layer of powdered flux material, the substrate comprising a composition beyond a zone of weldability defined on a graph of superalloys plotting titanium content verses aluminum content, wherein the zone of weldability is upper bounded by a line intersecting the titanium content axis at 6 wt. % and intersecting the aluminum content axis at 3 wt. %;

melting the powdered flux material and a surface region of the substrate comprising a surface opening crack with an energy beam to form a re-melted zone of the substrate underlying a layer of slag;

allowing the re-melted zone to cool and solidify under the layer of slag;

removing the slag; and

performing a post weld heat treatment on the substrate without forming reheat cracks.

10. The method of claim 9, wherein the energy beam is a laser beam.

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11. The method of claim 9, further comprising melting a filler material with the energy beam such that melted filler material is additive to the re-melted zone.

12. The method of claim 11, further comprising applying the filler material as powdered alloy material.

13. The method of claim 12, wherein a mesh size range of the powdered alloy material overlaps with a mesh size range of the powdered flux material.

14. The method of claim 11, further comprising applying the filler material as wire or strip material.

15. The method of claim 9, further comprising applying heat to the substrate prior to the step of melting.

16. A method comprising:

removing a superalloy hot gas path component from service in a gas turbine engine;

applying flux material to a surface of the component containing a surface opening defect;

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melting the surface of the component and the flux material with an energy beam to form a re-melted layer under a layer of slag without previously performing any subsurface crack cleaning step;

allowing the re-melted layer to cool under the slag;

removing the slag to reveal a repaired surface of the component without the defect, and

performing a heat treatment on the component after the step of removing the slag without forming reheat cracks.

17. The method of claim 16, wherein the substrate comprises a composition beyond a zone of weldability defined on a graph of superalloys plotting titanium content verses aluminum content, wherein the zone of weldability is upper bounded by a line intersecting the titanium content axis at 6 wt. % and intersecting the aluminum content axis at 3 wt. %.

18. The method of claim 16, further comprising applying heat to the surface of the component prior to the step of melting.

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